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DESCRIPTION

RARE EARTH OXIDE SUPERCONDUCTOR AND PROCESS FOR PRODUCING THE
SAME

Technical Field

The present invention relates to an oxide superconductor and a method for producing the same, and, in particular, to a tape-formed rare earth oxide superconductor suitable for applications to such instruments as superconductive magnets and superconductive cables, and an improvement in a method for producing the same.

Background Art

Since a rare earth 123 oxide superconductor (RE-123 superconductor) excels in magnetic-field properties at the temperature of liquid nitrogen as compared with a Bi type superconductor, it can realize a practical high critical current density (J_c) in a high magnetic field. Accordingly, if the practical use of a wire material thereof is achieved, in addition to excellent properties at high temperature regions, there is an economically extraordinary advantage because a production method without using silver, which is noble metal, is possible, and, since liquid nitrogen can be used as a refrigeration medium, cooling efficiency is enhanced to

several tens to several hundreds times. The result makes it possible to apply a superconductive wire material to instruments, to which the application has been conventionally impossible from the economic perspective. Thus, significant expansion of the application and market of superconductive instruments is foreseen.

RE-123 super conductors (particularly, Y-123 super conductor, Y:Ba:Cu = 1:2:3 at mole ratio) has an orthorhombic crystal system. Therefore, in order to make them exert characteristics of the material in conductive characteristics, it is required not only to align CuO planes of the crystal, but also to align in-plane crystal orientations. The reason is that a slight misalignment of orientations creates a twin boundary to lower conductive characteristics.

A manufacturing method of a wire material of the Y-123 super conductor while enhancing in-plane alignment and aligning the in-plane orientation of crystals thereof has the same way as a manufacturing method of a thin film. That is, it is possible to enhance the in-plane alignment degree and orientation of crystals of Y-123 super conductor by forming an intermediate layer having been enhanced in the in-plane alignment degree and the orientation on a tape-formed metal substrate, and using the crystal lattice of the intermediate layer as a template.

Further, it has been proved that J_c of a superconductor

depends on crystallinity and surface smoothness of an intermediate layer, and that characteristics thereof changes sensitively and significantly according to conditions of an underlying layer.

With regard to production techniques of above-described biaxially aligned metal substrate in which an in-plane aligned intermediate layer is formed on a tape-formed metal substrate, there are known such methods as an SOE (Surface-Oxidation Epitaxy) method, an ISD (Inclined Substrate Deposition) method, an IBAD (Ion Beam Assisted Deposition) method and a RABiTS (Rolling Assisted Biaxially Textured Substrate) method, and there are many reports about Y-123 superconductive wire materials having a J_c of more than 10^6 A/cm² by forming an intermediate layer, whose in-plane alignment degree and orientation have been enhanced, on a non-aligned or aligned metal tape.

Among these, in formation of the intermediate layer in IBAD and RABiTS methods, a vacuum process based on a vapor phase method such as a PLD (Pulse Laser Deposition) method is used, and in the IBAD method, a combination of hastelloy/YSZ/Y₂O₃, and, on the other hand, in the RABiTS method, a combination of Ni/CeO₂/YSZ/CeO₂ or the like are generally employed as a biaxially aligned metal substrate, which have such advantage that they can give a dense and smooth intermediate layer film (for example, refer to Non-patent Document 1).

There are many examinations about materials for the intermediate layer. Among these, a CeO_2 intermediate layer is known as one of the best intermediate layers since it has a good consistency of crystal lattice with a YBCO layer (Y-Ba-Cu-O superconductive layer) and a low reactivity with a YBCO layer, and many results are reported.

As described above, CeO_2 has excellent properties as an intermediate layer when forming a YBCO layer on a metal substrate. However, since a CeO_2 film is apt to easily crack because of differences in thermal expansion and the like from a metal substrate and can not be formed in a thick film, an intermediate layer of YSZ (yttrium-stabilized zirconia) or the like must be interposed between a CeO_2 film and a metal substrate to form a YBCO layer thereon, as is the case for the RABiTS method.

In order to solve the problem, the present inventors formed an intermediate layer composed of a cerium-based oxide incorporating cerium with 1 type or 2 or more types of rare earth elements, for example, of a solid solution generated between CeO_2 and Gd_2O_3 on a metal substrate by an MOD (Metal Organic Deposition Processes) method, and formed a rare earth oxide superconductive layer (RE superconductive layer) on the intermediate layer, thereby succeeded in forming a RE superconductive layer excelling in superconductive properties on an intermediate layer that can prevent cracking, excels in

crystallinity and surface smoothness, and is capable of low temperature synthesis (refer to Japanese Patent Application Nos. 2003-129368 and 2003-129369).

However, it was revealed that a YBCO layer formed on the intermediate layer by an MOD method shows a critical temperature (T_c) lower than the T_c of 90K fundamentally belonging to a YBCO layer by around 10K.

This is attributable to diffusion of a Ni element in the substrate into the superconductive layer, thereby substituting for a Cu element constituting the RE-123 super conductor.

Therefore, it was revealed that, although an intermediate layer excelling in lattice consistency, that is, crystallinity and surface smoothness and being capable of preventing cracking can be obtained when forming an intermediate layer of cerium-based oxide incorporating cerium with 1 type or 2 or more types of rare earth elements, for example, Ce-Gd-O by an MOD method as an intermediate layer on a metal substrate, there is a room for improvement as regarding an inhibiting effect on diffusion of an element constituting a substrate.

[Non-patent Document 1] A. Goyal et al., Physica C, 357-360 (2001) 903.

Disclosure of the Invention

The present invention has been achieved in order to solve the above problem, and it has for its objects to provide a rare earth oxide superconductor (RE superconductor) in which an intermediate layer that is capable of preventing cracking and preventing an element constituting a substrate from diffusing into a superconductive layer and excellent in an in-plane alignment degree and crystallinity such as orientation and smoothness is calcined on a metal substrate, and a RE superconductive layer excelling in J_c and T_c is formed on the intermediate layer; and a method for producing the same.

Brief Description of the Drawings

Fig. 1 is a cross-sectional view of the rare earth oxide superconductor according to Example of a first embodiment of the present invention.

Fig. 2 is a cross-sectional view of the rare earth oxide superconductor according to Example of a second embodiment of the invention.

Fig. 3 is a cross-sectional view of the rare earth oxide superconductor according to Example of a third embodiment of the invention.

Fig. 4 is a graph showing the distribution of Ni element at the cross-section of the rare earth oxide superconductor of Example of the second embodiment of the invention.

Fig. 5 is a graph showing the distribution of Ce element

and Gd elements at the cross-section of the rare earth oxide superconductor of Example of the second embodiment of the invention.

Fig. 6 is a graph showing the distribution of Nb element at the cross-section of the rare earth oxide superconductor of Example of the second embodiment of the invention.

Best Mode for Carrying Out the Invention

In order to accomplish the object, for a RE superconductor as a first embodiment of the present invention, it is intended to form, on the surface of a metal substrate, an intermediate layer constituted by sequentially forming a first intermediate layer composed of cerium and a solid solution formation element capable of forming a solid solution with cerium, and a second intermediate layer composed of a charge compensation element capable of compensating for a charge mismatch attributable to a difference between the electron valences of respective ions of cerium and the solid solution formation element capable of forming the solid solution with cerium, and then form a RE superconductive layer thereon.

The structure of the superconductor in this case is represented, for example, as Ni substrate/ Ce-Gd first intermediate layer/ Ce-Nb second intermediate layer/ YBCO layer, as shown in Fig. 1.

Further, for a RE superconductor as a second embodiment of the invention, it is intended to form the second intermediate layer in the first embodiment with cerium, the solid solution formation element and the charge compensation element.

The structure of the superconductor in this case is represented, for example, as Ni substrate/ Ce-Gd first intermediate layer/ Ce-Gd-Nb second intermediate layer/ YBCO layer, as shown in Fig. 2.

Furthermore, for a RE superconductor as a third embodiment of the invention, it is intended to form an intermediate layer composed of cerium, cerium and the solid solution formation element, and the charge compensation element on the surface of a metal substrate, and then form a RE superconductive layer on the intermediate layer.

The structure of the superconductor in this case is represented, for example, as Ni substrate/ Ce-Gd-Nb intermediate layer/ YBCO layer, as shown in Fig. 3.

In the aforementioned inventions, as the solid solution formation element, any one type or 2 or more types of rare earth elements including Y, Nd, Sm, Gd, Eu, Yb, Ho, Tm, Dy, La and Er are used; and, on the other hand, as the charge compensation element, any one type or 2 or more types of elements including Bi, Nb, Sb, Ta and V are used.

The aforementioned RE superconductors are produced according to following methods for producing a RE

superconductor, which are a fourth to a sixth embodiments of the invention.

That is, in a method for producing a RE superconductor as a fourth embodiment of the invention, it is intended to form a first coated layer on the surface of a metal substrate by applying a mixed solution composed of an organometallic acid salt of cerium and an organometallic acid salt or salts of any 1 type or 2 or more types of solid solution formation elements capable of forming a solid solution with cerium selected from (Y, Nd, Sm, Gd, Eu, Yb, Ho, Tm, Dy, La and Er), which was then subjected to preliminary calcination, followed by forming a second coated layer on the first coated layer by applying a mixed solution composed of an organometallic acid salt of cerium and an organometallic acid salt or salts of any 1 type or 2 or more types of charge compensation elements capable of compensating for a charge mismatch attributable to a difference between the electron valences of respective ions of cerium and the solid solution formation element selected from (Bi, Nb, Sb, Ta and V), subsequently carrying out heat treatment in a reducing atmosphere under a temperature ranging from 900 to 1200°C and a pressure ranging from 0.1 Pa to below atmospheric pressure to form an intermediate layer, and then forming a RE superconductive layer on the intermediate layer.

Further, a method for producing a RE superconductor as a fifth embodiment of the invention is intended to form the

second coated layer in the fourth embodiment by coating a mixed solution composed of an organometallic acid salt of cerium, an organometallic acid salt or salts of 1 type or 2 or more types of solid solution formation elements selected from the solid solution formation element, and 1 type or 2 or more types of organometallic acid salts of the charge compensation element.

Furthermore, a method for producing a RE superconductor as a sixth embodiment of the invention is intended to form an intermediate layer on the surface of a metal substrate by coating a mixed solution composed of an organometallic acid salt of cerium, cerium and an organometallic acid salt or salts of 1 type or 2 or more types of the solid solution formation element, and an organometallic acid salt or salts of 1 type or 2 or more types of the charge compensation element, and carrying out heat treatment in a reducing atmosphere under a temperature ranging from 900 to 1200°C and a pressure ranging from 0.1 Pa to below atmospheric pressure.

According to the RE superconductor and the method for producing the same of the invention, by forming an intermediate layer on a metal substrate while using cerium (organometallic acid salt), a solid solution formation element (organometallic acid salt) capable of forming a solid solution with cerium, and a charge compensation element (organometallic acid salt) capable of compensating for a charge mismatch attributable to

a difference between electron valences of cerium and the solid solution formation element, and forming a RE superconductive layer on the intermediate layer, it is possible to form an intermediate layer capable of preventing cracking in the intermediate layer and excelling in crystallinity such as an in-plane alignment degree and orientation, and surface smoothness on a metal substrate, and to form a RE superconductive layer excelling in J_c and T_c on the intermediate layer while inhibiting diffusion of an element constituting the substrate.

In the RE superconductor and the method for producing the same according to the invention, an intermediate layer and a rare earth oxide superconductive layer are formed on the surface of a metal substrate, wherein the intermediate layer is required to have a low reactivity with the superconductor, and a small percentage of difference in crystal lattice distances (misfit), and further a function to prevent diffusion of an element of the underlying layer. As a crystalline structure appropriate for the intermediate layer from this viewpoint, any of such crystalline structures as fluorite structure, rare earth-C structure and pyrochlore structure is selected. Further, by forming a solid solution of Ce and a rare earth element, cracking of a conventional CeO_2 single layer can be prevented.

In this case, the misfit between the lattice constant

3.88 Å of the a-axis of an Y-123 superconductor crystal and the crystal lattice of the oxide becomes 8% or less, but the misfit varies depending on the composition and is, when possible, desirably 1% or less.

It is desirable that the content of the solid solution formation element in the first intermediate layer of the first and second embodiments of the invention, the content of the charge compensation element in the second intermediate layer of the first embodiment, the total content of the solid solution formation element and the charge compensation element in the second intermediate layer of the second embodiment or the total content of the solid solution formation element and the charge compensation element in the intermediate layer of the third embodiment are 5 to 60 mol% in terms of the metal content, respectively.

It is also desirable that the content of the solid solution formation element in the first coated layer of the fourth and fifth embodiments, the content of the charge compensation element in the second coated layer of the fourth embodiment, the total content of the solid solution formation element and the charge compensation element in the second coated layer of the fifth embodiment or the total content of the solid solution formation element and the charge compensation element in the intermediate layer of the sixth embodiment are, in the same way, 5 to 60 mol% in terms of the

metal content, respectively.

Because, in the embodiments, when the content of the solid solution formation element in an intermediate layer or a coated layer of a single layer is less than 5 mol%, the effect on preventing cracking is small; and when the content is more than 60 mol%, the effect as an intermediate layer becomes small because it has an enhanced reactivity with the YBCO layer and the metal substrate. In particular, it is preferable that the addition content of the solid solution formation element is 5 to 40 mol% in terms of the metal content. The solid solution formation element is a rare earth element having a charge of +3, which forms a solid solution with Ce of CeO_2 to fulfill a function of lowering the melting point or crystallization temperature in a reaction treatment of crystallization, as well as prevents cracking.

Accordingly, it is desirable that a charge compensation element in an intermediate layer or a coated layer of a single layer is also 5 to 60 mol% in terms of the metal content. The charge compensation element has a charge of +5, and compensates for a charge mismatch created due to a difference between the electro valences of respective ions of Ce and the solid solution formation element and inhibits an ion diffusion in the oxide film.

Further, when the solid solution formation element and the charge compensation element coexist in an intermediate

layer or a coated layer of a single layer, a mole ratio of the charge compensation element to the solid solution formation element is desirably charge compensation element/ solid solution formation element ≤ 1.2 .

When the intermediate layer is formed in 2 layers, it is necessary to carry out preliminary calcination after coating the first layer, which may be subjected to normal calcination (crystallization heat treatment).

The intermediate layer is formed by calcination in a reducing atmosphere under a pressure ranging from 0.1 Pa to below atmospheric pressure and a temperature ranging from 900 to 1200°C, and, subsequently, a RE superconductive layer is formed on the intermediate layer. In this case, it is more preferable to form the intermediate layer by calcination in a reducing atmosphere under a pressure ranging from 10 to 500 Pa and a temperature ranging from 950 to 1150°C.

Calcination carried out when the pressure is less than 0.1 Pa results in crystallization before epitaxial growth of the film to significantly lower the alignment of an intermediate layer. In particular, calcination in a range of 10 to 500 Pa can lower a heat treatment temperature at the crystallization to make calcination of an intermediate layer at a low temperature of 1000°C or less possible.

In addition, because, when the calcination temperature of an intermediate layer is a temperature less than 900°C, it

becomes difficult to obtain a biaxially aligned film; and a temperature more than 1200°C results in decomposition of a film at calcination making it difficult to give a targeted oxide.

With regard to a method for producing an intermediate layer, use of many film-forming methods via a liquid phase process is possible, but it is preferable to employ an MOD method from the point of easiness of production and production speed. The MOD method is known as a method of carrying out production by a non-vacuum process. In the case of the present invention, a mixed solution of trifluoroacetate (TFA salt), octylate, naphthenate or the like of cerium, a solid solution formation element and/or a charge compensation element that constitute an intermediate layer at a predetermined mole ratio is coated on a support, which is then subjected to preliminary calcination.

By carrying out calcination of an intermediate layer formed by an MOD method under a reduced pressure described above, crystallization temperature can be lowered, and calcination of an intermediate layer at a low temperature becomes possible. When a metal tape is used as a substrate, this is effective in such point that a diffusion speed of an underlying layer metal element into the inside of the intermediate layer is lowered.

As a metal substrate, such metal tapes can be used as biaxially aligned metal tapes composed of Ni, Ag or alloys of

these elements, for example Ni-V or Ni-W alloy, or metal tapes composed of non-aligned Ni, Ag or alloys of these elements, or heat-resistant alloys such as SUS, hastelloy or Inconel.

When forming an intermediate layer directly on a metal substrate composed of Ni, Ni-based alloy or the like, as an atmosphere at calcination, use of a reducing atmospheric gas prepared by adding 0.1 to 10% of H₂ to Ar or N₂ gas is preferable. Because when a film is formed in an atmosphere of H₂ concentration of less than 0.1%, NiO generates on the surface of Ni to significantly block epitaxial growth of the intermediate layer; and, since reducing power of the gas becomes too strong, that of more than 10% results in not giving a targeted oxide.

As described above, it is also possible to dispose an alignment-controlling and diffusion-preventing layer of 0.2 μm or less formed by using a vapor phase process such as a PLD method or a sputtering method between an intermediate layer formed by an MOD method and a metal tape. Further, it is also effective to carry out film-forming a CeO₂, Ce-RE-O film of 0.2 μm or less on an intermediate layer manufactured by an MOD method as a CAP layer.

The repetition number of coating a precursor film of an intermediate layer on a metal substrate formed by an MOD method is not restricted in any way and, in order to obtain a designed film thickness, it is also possible to employ a procedure in

which a coating-preliminary calcination (drying) treatment is carried out more than once. In other words, independently of the repetition number of coating, it is sufficient when the charge balance in the formed precursor film of an intermediate layer is compensated for.

With regard to a method for forming a superconductive layer on the intermediate layer thus formed on a metal substrate, it is possible to form it according to various methods including physical evaporation methods such as a pulse laser deposition (PLD) method and an e-beam evaporation, gas phase processes such as a chemical evaporation method including a chemical vapor deposition (CVD) method, and a film-forming method via a liquid phase process such as an MOD method as is the case with an intermediate layer.

In particular, the intermediate layer according to the present invention is remarkably effective in a method, among the methods for forming a superconductive layer, in which film-forming of a precursor of TFA subjected to preliminary calcination (TFA-MOD method), or film-forming of a F-containing precursor by an e-beam or PLD method (ex-situ method) on a tape surface is carried out followed by normal calcination to form a YBCO film. In these processes, since the precursor film contains F and water vapor is used at calcination, HF generates at preliminary calcination and normal calcination, and, accordingly, acid resistance is

required for an intermediate layer. On this occasion, since the intermediate layer according to the present invention is of a Ce-based oxide, it excels in acid resistance.

Further, since the intermediate layer in the present invention can prevent cracking, it can be formed in an increased thickness, which makes it unnecessary to form a multi-layer structured intermediate layer as is the case for the RABiTS method, and makes it possible to form a rare earth oxide superconductive layer directly on a cerium-based oxide layer. (Formation of an intermediate layer and a YBCO layer according to an MOD method)

An aligned Ni substrate was used as a substrate, on which an intermediate layer film and a YBCO film were formed by an MOD method.

Solutions of various organometallic compounds having a metal concentration of 0.2 mol/L were prepared as raw materials, each of which was coated on a substrate having a size of 10 mm × 5 mm by a spin coating method to form a coated film. The rotation speed at that time was 3000 rpm. The substrate coated with the precursor film of an intermediate layer film was subjected to preliminary calcination and drying in an Ar-H₂ (2%) atmosphere, and further subjected to calcination in a reducing atmosphere to form an intermediate layer film on the substrate.

After forming the intermediate layer film on the

substrate, a YBCO layer was film-formed on the intermediate layer film by a TFA-MOD method, which was then subjected to a heat treatment of $750^{\circ}\text{C} \times 1$ hour in an oxidizing atmosphere to produce a superconductor.

Example 1

A mixed solution of naphthenates of $\text{Ce}:\text{Gd} = 7:3$ at mole ratio was used as a raw material solution, which was coated on a substrate, and then subjected to preliminary calcination at 300°C to form a first intermediate layer. Subsequently, on the first intermediate layer was coated a mixed solution of naphthenates of $\text{Ce}:\text{Nb} = 7:3$ at mole ratio, and then subjected to a heat treatment of $1000^{\circ}\text{C} \times 1$ hour in a reducing atmosphere of $\text{Ar}-2\%\text{H}_2$ at a pressure of 10 Pa to form an intermediate layer. On the intermediate layer, a YBCO layer was formed by a TFA-MOD method.

The YBCO film manufactured on the intermediate layer showed a critical temperature (T_c) of from 85 to 88K.

Example 2

A mixed solution of naphthenates of $\text{Ce}:\text{Gd} = 6:4$ at mole ratio was used as a raw material solution, which was coated on a substrate, and then subjected to preliminary calcination at 300°C to form a first intermediate layer. Subsequently, on the first intermediate layer was coated a mixed solution of

naphthenates of Ce:Gd:Nb = 6:2:2 at mole ratio, which was subjected to a heat treatment of $1000^{\circ}\text{C} \times 1$ hour in a reducing atmosphere of Ar-2%H₂ at a pressure of 20 Pa to form an intermediate layer. On the intermediate layer, a YBCO layer was formed by a TFA-MOD method.

The YBCO film manufactured on the intermediate layer showed a critical temperature (T_c) of from 85 to 88K.

Example 3

A mixed solution of naphthenates of Ce:Gd:Nb = 90:5:5 at mole ratio was used as a raw material solution, which was coated on a substrate, and then subjected to preliminary calcination at 300°C , and subsequently, a mixed solution of naphthenates of Ce:Gd:Nb = 70:20:10 at mole ratio was coated thereon, which was subjected to a heat treatment of $1000^{\circ}\text{C} \times 1$ hour in a reducing atmosphere of Ar-2%H₂ at a pressure of 500 Pa to form an intermediate layer. On the intermediate layer, a YBCO layer was formed by a TFA-MOD method.

The in-plane alignment of the obtained intermediate layer film slightly depended on the preliminary calcination condition and showed a FWHM ranging from 8 to 10 degrees. The YBCO film manufactured on the intermediate layer showed a critical temperature (T_c) of 88K.

Example 4

A mixed solution of naphthenates of Ce:Gd = 90:10 at mole ratio was used as a raw material solution, which was coated on a substrate, and then subjected to preliminary calcination at 300°C to form a first intermediate layer. Subsequently, on the first intermediate layer was coated a mixed solution of naphthenates of Ce:Gd:Nb = 70:20:10 at mole ratio, which was subjected to a heat treatment of 1000°C × 1 hour in a reducing atmosphere of Ar-2%H₂ at a pressure of 500 Pa to form an intermediate layer. On the intermediate layer, a YBCO layer was formed by a TFA-MOD method.

The YBCO film formed on the intermediate layer showed a critical temperature (T_c) of 88K and a critical current density (J_c) of 0.3 MA/cm². Further, the intermediate layer showed a FWHM of 7.6 degrees, and the YBCO layer showed a FWHM of 8.4 degrees, which means a good degree of alignment.

Distribution of elements added in the intermediate layer thus formed is shown in Figs. 4-6. Here, (A) denotes the position on the intermediate layer surface and (B) denotes the position on the Ni substrate surface.

As is clear from these drawings, by forming an intermediate layer having a two-layer structure of Ce-Gd/Ce-Gd-Nb, it is possible to prevent diffusion of a Ni element constituting a substrate into a YBCO layer, it becomes possible to give an intermediate layer and YBCO layer having a good degree of alignment, and, as the result, it is possible

to produce a superconductor having an excellent superconductive properties.

Comparative Example 1

A mixed solution of naphthenates of Ce:Gd:O = 60:20:20 at mole ratio was prepared and coated on a substrate, which was subjected to preliminary calcination at 300°C, and then subjected to normal calcination of 1000°C × 1 hour in a reducing atmosphere of Ar-2%H₂ at a pressure of 20 Pa.

A YBCO film formed on the intermediate layer by a TFA-MOD method showed a critical temperature (T_c) of 80K, which was lower than 90K as the physical property value of YBCO superconductor by 10K.

Comparative Example 2

A mixed solution of naphthenates of Ce:Gd:O = 90:5:5 at mole ratio was prepared and coated on a substrate, which was subjected to preliminary calcination at 300°C, and then subjected to normal calcination of 1000°C × 1 hour in a reducing atmosphere of Ar-2%H₂ at a pressure of 20 Pa.

A YBCO film formed on the intermediate layer by a TFA-MOD method showed a critical temperature (T_c) of 70K, which was lower than 90K as the physical property value of YBCO superconductor by 20K.

As is clear from the Examples and Comparative examples,

when Nb as a charge compensation element is not added to cerium, Tc is 70-80K; on the contrary, when Nb as a charge compensation element is added to Gd as a solid solution formation element, Tc is wholly improved such as 85-88K. This result is thought to be attributable to compensation of a deficiency of charge caused by addition of a solid solution formation element to lower ion diffusion in the intermediate layer, thereby preventing a Ni element from diffusing into a superconductive layer.

Industrial Applicability

The rare earth oxide superconductor according to the present invention is useful for a tape-formed rare earth oxide superconductor suitable for use in instruments such as superconductive magnets and superconductive cables.